

SYNTHESIS OF (Ba,Pb)TiO₃ SOLID SOLUTION HAVING NO COMPOSITIONAL FLUCTUATION

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A solid solution of (Ba,Pb)TiO₃ without compositional fluctuation has not been obtained by an ordinary solid-solid reaction of BaCO₃, PbO and TiO₂. This was confirmed through the measurements of $\Delta d/d$ using X-ray diffraction. A new method of preparing (Ba,Pb)TiO₃ without compositional fluctuation was developed.

A number of compounds with perovskite structure exhibit ferroelectricity. They are widely used for various electric devices. Barium titanate, one of the most important ferroelectric materials for applications, has relatively low Curie temperature (120°C). To shift the Curie point to higher temperatures, PbTiO₃ is added, yielding (Ba,Pb)TiO₃ solid solutions(BPT). Generally solid solution tends to have a compositional fluctuation. Sometimes the compositional fluctuation is harmful because of causing a diffuse ferroelectric-paraelectric phase transition. In addition, the compositional fluctuation complicates a morphotropic phase boundary¹⁻²). A chemical preparation in liquid phase is effective to eliminate the compositional fluctuation. But only a few perovskite compounds can be prepared by a simultaneous precipitation because it is often difficult to find a common condition precipitating all the constituent cations. Most A-site cations are hard to be precipitated by a simple hydroxidation. Moreover, some anion (like Cl⁻) in a stable solution of cations (ex. Ti⁴⁺, Zr⁴⁺) makes certain cation (like Pb²⁺) insoluble. For example, it is impossible to make a mixed solution of TiCl₄ and Pb(CH₃COO)₂. Consequently there can be a difficulty in making a solution consisting of all the constituent cations. Let us note the following two facts: (A) A solid-solid reaction of PbO (A-site component) and TiO₂ (B-site component) forms PbTiO₃ having no compositional fluctuation³⁻⁴). (B) Pb(Zr,Ti)O₃ prepared by a solid-solid reaction among PbO, TiO₂, and ZrO₂ has a compositional fluctuation⁵⁻⁶). Considering these facts, it is interesting to examine the compositional fluctuation of a solid solution prepared by a solid-solid reaction between a coprecipitate of A-site constituents and that of B-site constituents. If a solid solution having no compositional fluctuation be obtained by this method, it will be very useful because we need not coprecipitate all the constituent cations. We have already examined a particular case¹), i.e. a reaction between PbO (A-site component) and a coprecipitate of Zr⁴⁺ and Ti⁴⁺ (B-site constituents). Pb(Zr,Ti)O₃ prepared by this method has no compositional fluctuation. This example is a B-site substituted solid solution. Effectiveness

of the method of solid-solid reaction between a coprecipitate of A-site constituents and a B-site component has been left unexamined. Consequently, $(A_I, A_{II})BO_3$ having no compositional fluctuation has not been obtained thus far. To get $(Ba, Pb)TiO_3$ having no compositional fluctuation is important not only for a preparation of excellent dielectrics but for investigating an effect of vaporization of Pb on the compositional fluctuation. It was difficult to precipitate Pb^{2+} and Ba^{2+} simultaneously and homogeneously. Actually we tried to precipitate them as an oxalate, but failed in obtaining a homogeneous precipitate⁷⁾. We wish to report on a new method of preparing $(Ba, Pb)TiO_3$ solid solution having no compositional fluctuation.

$BaCO_3$ (99.999%, Johnson Matthey Chemicals Ltd.), PbO (99%, Junsei Chemical Co. Ltd.) and TiO_2 (99.98%, TP-3, Fuji Titanium Co. Ltd.) were thoroughly mixed in a

mortar with a pestle. The mixture was pressed into a powder compact, placed in an alumina crucible and fired at desired temperature. In this procedure the specimen was covered with another smaller alumina crucible and a powder of the same composition as the sample was packed over the latter crucible to supply PbO vapor around the sample.

A different method was undertaken to get the solid solution. A mixed aqueous solution of $Pb(CH_3COO)_2 \cdot 3H_2O$ (Guaranteed reagent, Wako Pure Chemical Industries. Ltd.) and that of $Ba(CH_3COO)_2 \cdot H_2O$ (Guaranteed reagent, Kanto Chemical Co. Ltd.) were prepared, dropped into a solution of 6M $(NH_4)_2CO_3$ (Guaranteed reagent, Kanto Chemical Co. Ltd.), to get a white precipitate. This was washed and dried at 200°C. The X-ray diffraction pattern showed the dried material to be a single phase of $(Ba, Pb)CO_3$ (aragonite structure). The mole ratio of Pb/Ba for typical 3 specimens was determined by fluorescence X-ray analysis (Rigaku Denki Co. Ltd., Type-3064), and compared with that of the starting powders (Table 1). The resultant values are very close to the desired ones indicating that the foregoing

Table 1 Chemical analysis of the co-precipitates

Sample	Pb/(Pb+Ba)	
	desired	analysed
A	0.7	0.703
B	0.7	0.696
C	0.7	0.696

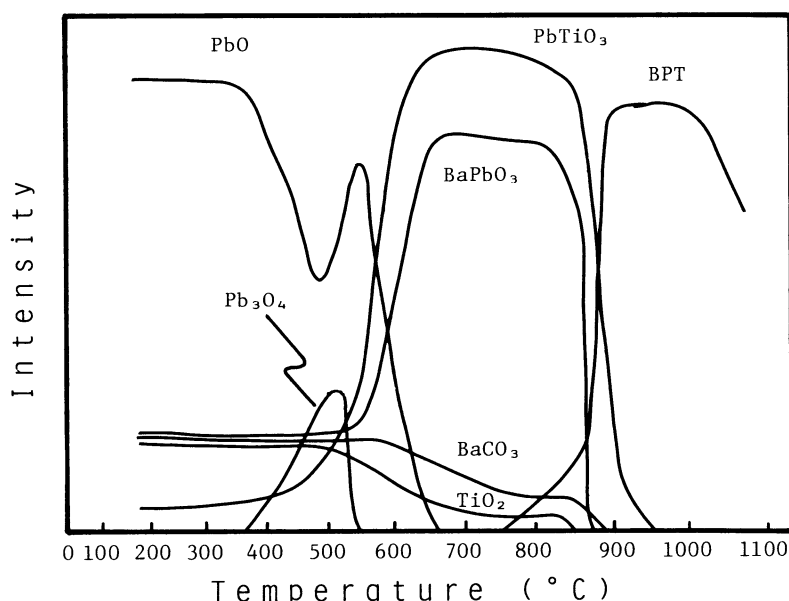


Fig. 1 Reaction processes for BPT prepared by a wet-dry combination technique.

scanning region: 23-25° and 25-35°

heating rate: 5°C/min

chemical precipitation proceeded quantitatively. The precipitate was fired at 700°C for 1 h and then mixed dry with TiO₂. Experimentally, this firing temperature and period was the most favorable condition. Details on this condition will be reported near future. The resultant mixture was fired at a desired temperature to get a BPT solid solution. This procedure is hereafter called "wet-dry combination technique". A high temperature X-ray diffractometer (JEOL) was used for tracing the formation process of the material (Fig. 1). The characteristic peaks of interest were scanned by oscillating in the ranges of $2\theta=25-35^\circ$ and $23-25^\circ$ with the radiation of CuK α and a temperature elevation speed of 5°C/min. The starting materials were PbO, BaCO₃, BaPbO₃ and TiO₂ where PbO and BaCO₃ were formed as a result of the decomposition of (Ba,Pb)CO₃ at 700°C firing. After PbO transformed into Pb₃O₄ in the vicinity of 380°C, the resultant Pb₃O₄ converted again to PbO in the vicinity of 500°C⁸⁾. Above 550°C, BaPbO₃ was formed by the reaction between these PbO and BaCO₃, and at the same time PbTiO₃ appeared in accordance with disappearance of TiO₂. BPT appeared rapidly at about 900°C, and at the same time PbTiO₃, BaPbO₃ and TiO₂ disappeared. It is thought that a gradual decrease of the BPT peak intensity with temperature above 1000°C corresponds to an evaporation of PbO in the material, since the reaction was carried out in an open atmosphere. This fact indicates that samples should be fired in a PbO vapor atmosphere to get a stoichiometric material as were actually done in the preparation of BPT.

The system of BaTiO₃-PbTiO₃ makes a complete solid solution over a whole range, and exhibits a tetragonal crystal system. As reported elsewhere⁹⁾, the axis ratio, c/a, of PbTiO₃ is largest among the compounds with perovskite structure. With an increase in x of (Ba_xPb_{1-x})TiO₃, the c-axis decreased and the a-axis increased, while c/a decreased. The compositional parameter x dependence of c-axis was somewhat larger than that of a-axis. The fluctuation of lattice spacing of a material is generally determined from the slope of the plots of $\beta \cdot \cos\theta$ vs. $\sin\theta$ ¹⁰⁾, where β is the net X-ray diffraction line broadening originating from the material only. The plots of $\beta \cdot \cos\theta$ vs. $\sin\theta$ for (Ba_{0.3}Pb_{0.7})TiO₃ prepared by the solid-solid reaction (1200°C, 1 h) are shown

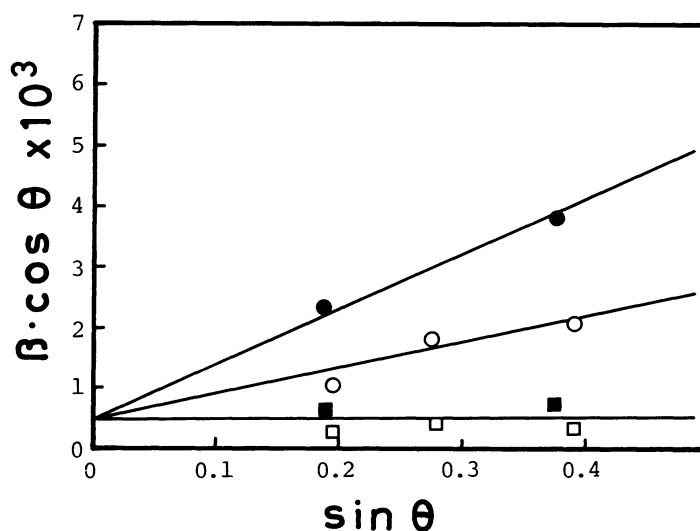


Fig. 2

Plots of $\beta \cdot \cos\theta$ vs $\sin\theta$ for (Ba_{0.3}Pb_{0.7})TiO₃
For sample prepared by a solid-solid reaction

○:hk0 diffraction
●:00l diffraction

For sample prepared by a wet-dry combination technique

□:hk0 diffraction
■:00l diffraction

for $hk0$ and $00l$ diffraction of the material in Fig. 2. As seen there, both slopes for $hk0$ and $00l$ peaks were positive and the slope for the diffraction of the latter was larger than that for the former. These slopes is interpreted in terms of compositional fluctuation of the material. Remarks is necessary that the same type of plots for pure $BaTiO_3$ and $PbTiO_3$ prepared by the ordinary dry method gave zero gradients, from which one can not expect the compositional fluctuation. The plots of $\beta \cdot \cos\theta$ vs. $\sin\theta$ for $(Ba_{0.3}Ti_{0.7})O_3$ prepared by the wet-dry combination technique are also shown in Fig. 2. Both slopes for $hk0$ and $00l$ diffractions are zero, indicating that the material has no compositional fluctuation. The lack of the compositional fluctuation in this material may be originated from their homogeneous distribution in the mixed solution of Pb^{2+} and Ba^{2+} . As mentioned before, the result of high temperature diffraction suggests the formation of $BaPbO_3$ and $PbTiO_3$ as intermediate products. It is noted that such a coexistence of $PbTiO_3$ and $BaTiO_3$ can eventually go to homogenization of Pb^{2+} and Ba^{2+} ; this implies that the two by-products were homogeneously-mixed considerably. It is concluded that the method of obtaining A-site substituted solid solution by a solid-solid reaction between a coprecipitate of A-site constituents and a B-site component is successful. The ferroelectric material prepared by this method may be very useful in their applications. It would provides a preliminary step to get $(A_I, A_{II})(B_I, B_{II})O_3$ type perovskite compound without compositional fluctuation by a similar method.

References

- 1) K. Kakegawa, J. Mohri, K. Takahashi, and S. Shirasaki, *Solid State Commn.*, 24, 769 (1977).
- 2) K. Kakegawa, J. Mohri, S. Shirasaki, and K. Takahashi, *J. Amer. Ceram. Soc.*, 65, October (1982) in press.
- 3) K. Kakegawa, J. Mohri, S. Shirasaki, H. Yamamura, and K. Takahashi, *Nippon Kagaku Kaishi*, 1974, 10.
- 4) K. Kakegawa, J. Mohri, H. Yamamura, and S. Shirasaki, *Jpn. J. Appl. Phys.*, 12, 1821 (1973).
- 5) K. Kakegawa, J. Mohri, K. Takahashi, H. Yamamura, and S. Shirasaki, *Nippon Kagaku Kaishi*, 1976, 717.
- 6) K. Kakegawa, K. Watanabe, J. Mohri, H. Yamamura, and S. Shirasaki, *Nippon Kagaku Kaishi*, 1975, 413.
- 7) J. Mohri, K. Kakegawa, H. Ishiguro, K. Takahashi, and S. Shirasaki, Presented in The 19th Meeting on Ceramic Science (1981), Nagoya.
- 8) J. W. Mellor, "A Comprehensive Treatise on Inorganic and Theoretical Chemistry", Vol. VII, Longmans, Green and Co Ltd, London (1963) p.674.
- 9) G. Shirane, S. Hoshino, and K. Suzuki, *J. Phys. Soc. Jpn.*, 5, 453 (1950).
- 10) A. R. Stokes, "X-Ray Diffraction by Polycrystalline Materials, Chapter 17", Chapman & Hall Ltd., London (1960).

(Received July 14, 1982)